

(373) On the Melting of Scrap and Sponge Iron (2)\*1,2)

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1. Introduction

The melting of pure solids in their own melt can be described by simple formulas (1,2). In the more applied case of melting and dissolution in a multi-component-system heat- and mass transport are studied. For the melting of sponge iron in a Fe-C alloy the influence of the carbon content on the transport phenomena is explained thermodynamically.

2. Some principals of melting

A solid that is immersed in a melt will form a shell.

If two samples of same mass and surface area but different heat conductivity are dipped into a melt, a thick shell will freeze on the sample with high heat conductivity. But actually this formation of a thick shell improves melting as can be seen at  $t = t_2$  in Fig. 1.

3. Melting in a binary system

During the melting process in a binary system, heat and mass have to be transported. Assuming thermodynamic equilibrium at the phase boundary, no heat and mass transport inside the sample and corresponding heat and mass transfer, a connection between carbon concentration  $x_1$  and temperature  $T_p$  at the phase boundary shown in Fig.2 can be derived.  $x_1$  and  $T_p$  are estimated from the liquidus line of the phase diagram. Furthermore it is assumed that the equilibrium state at the melting front is reached without any interference by the concentration  $x_c$  of the solid(2,3).

4. Heat- and mass transfer coefficients  $\alpha$  and  $\beta$  of sponge iron in an Fe-C alloy

The melting time  $t_E$  of pressed iron powder samples of mass  $M_0$ , surface  $F_C$  and initial temperature  $T_C$  has been reported to depend highly on the carbon content  $x_B$  of the bath with temperature  $T_B$  (Fig.3).

Using the formulas mentioned above, one can calculate the heat- and mass-transfer coefficients  $\alpha$  and  $\beta$ . It can be seen that the heat- and mass-transfer coefficients do not depend on the carbon content of the bath (Fig.4).

5. Conclusion

The great influence of the carbon on the melting time is therefore not a reason of kinetics, as some authors have assumed, but of thermodynamics.

References:

- 1) Friedrichs, H.A. et.al.: Tetsu-to-Hagané 68(1982), p. S-234
- 2) Rademacher, P.K.: Dr.-Ing.thesis, RWTH Aachen (1982) to be published
- 3) Friedrichs, H.A. et.al.: 5th Germany-Japan Seminar (1982) Duesseldorf, VDEh, p.161/78
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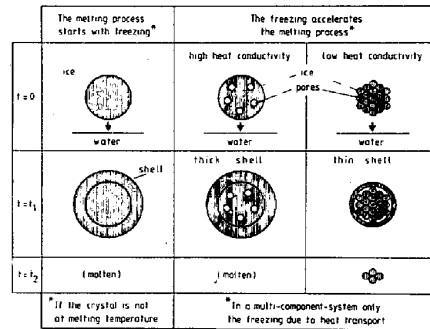


Fig.1: Some principles of melting

$$\frac{\ln(1 - \frac{x_B - x_1}{x_1 - x_c})}{\ln(1 + \frac{\Delta H_B}{\Delta H_P})} = \left(\frac{\alpha}{\beta}\right)^{1/2}$$

$$\alpha = \frac{1}{t_E} \frac{c_p M_0}{F_C} \frac{1}{\ln(1 + \frac{\Delta H_B}{\Delta H_P})} \quad \beta = \frac{\alpha}{c_p \rho_1} \left(\frac{\rho_1}{\rho_1}\right)^{1/2}$$

$$\Delta H_B = c_p(T_B - T_p) \quad \Delta H_P = H(T_p) - H(T_c) + \Delta H_m$$

Data used for the calculation of the transfer coefficients shown in Fig.4:

- Thermal diffusivity of the liquid:  $a_1 = 0.02 \text{ cm}^2/\text{s}$
- Diffusion coefficient of carbon:  $D/(cm^2 \cdot s) = 11 \cdot 0.26 \frac{x_1 - x_B}{296} \cdot 10^{-4}$
- Specific heat of the solid:  $H(T_p) - H(T_c) = (1740 + 0.71 T_p - 14.20 T_p^2) \cdot 10^{-4}$
- Specific heat capacity of the liquid:  $c_p = 0.71 \text{ J/(g K)}$
- Heat of fusion:  $\Delta H_m = 250 \text{ J/g}$
- Density of the liquid:  $\rho_1 = 7.1 \text{ g/cm}^3$

Fig.2: Formulas for the concentration  $x_1$  and temperature  $T_p$  at the phase boundary and for the transfer coefficients  $\alpha$  and  $\beta$ .

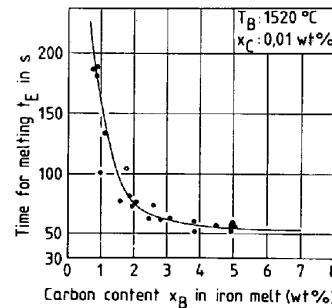


Fig.3: Reported values for the melting time  $t_E$  of pressed iron powder slabs (4)

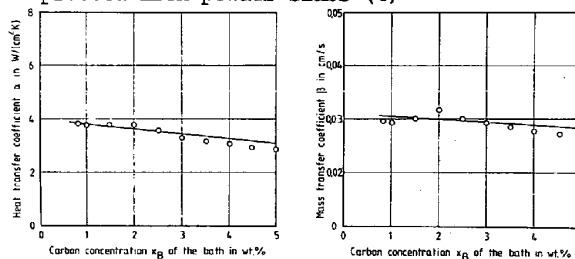


Fig.4: Calculated heat and mass transfer coefficients  $\alpha$  and  $\beta$  according to Fig.3 as functions of the carbon content  $x_B$  of the bath